of the SiO₄ unit are linked to Si and the fourth is linked to Al. The shifts observed for the inverted SiO₄ tetrahedra are in the range normally observed for Q⁴(1Al) sites in zeolites (-96 to -107 ppm).⁹ Deconvolution of the ²⁹Si MAS NMR spectrum for I indicates that 40% of the SiO₄ tetrahedra in the layers have reacted. Presumably, the Al centers to which the layers couple retain the octahedral environment of a Keggin-like structure, because there is little change in the ²⁷Al MAS NMR spectrum upon layer cross-linking.

The reaction of SiO_4 units in alumina pillared fluorhectorite but not alumina pillared montmorillonite or Laponite apparently is related to the presence of F in the former clay which labilizes Si-O bonds and promotes coupling. In montmorillonite and Laponite the layers are unreactive and pillaring presumably involves van der Waals interactions or, more likely, simple dative bond formation between the layer oxygens and coordinatively unsaturated sites on the alumina aggregates.

Cross-linking mechanisms that involve structural rearrangements of the host layers should be of fundamental importance in understanding the thermal stability and catalytic properties of the pillared clay. Future studies will relate changes in structural properties with differences in stability and catalytic activity.

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Highly Conductive Metallomacrocyclic Assemblies. Synthesis via Electrocrystallization and Single-Crystal Properties of a Phthalocyanine "Molecular Metal" without Halogen Counterions

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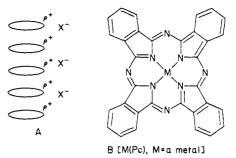
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The structural, electrical, magnetic, and optical properties of many low-dimensional molecular conductors¹ are a sensitive function of the interactions taking place between the band-forming molecular arrays and charge-compensating, off-axis counterions (e.g., A). To varying, largely unquantified degrees, such in-



teractions likely influence crystal structure,^{1,2} band-filling (ρ^+) ,¹⁻³ Coulombic screening,^{1,3,4} and charge and spin density wave phenomena,^{1,5} as well as the onset of superconductivity.^{1,5} Despite recent advances that have been achieved in modifying the M and Pc components of phthalocyanine (B) molecular conductors,^{2a,6} materials having counterions other than halogens (usually I₃⁻)^{2a} are conspicuous in their paucity.⁷ We communicate here the single-crystal synthesis via high-temperature electrocrystallization⁸ and some of the interesting properties of the first non-halogenated phthalocyanine "molecular metal", Ni(Pc)(BF₄)_{0,33} (1).⁹

Because of severe solubility limitations, fragile, needlelike, golden crystals of 1 were necessarily grown at high temperatures (>100 °C), galvanostatically (0.5 μ A, Pt wire anode) from solutions of rigorously purified Ni(Pc), Bu₄N⁺BF₄⁻, and 1-chloronaphthalene under anaerobic/anhydrous conditions. Larger quantities in powder form (identical spectroscopically, analytically, and diffractometrically) can be prepared from Ni(Pc) and NO⁺BF₄⁻ in CH₂Cl₂.⁹ Preliminary work^{10a} indicates that this electrochemical approach is generalizable to a wide range of counterions (e.g., PF₆⁻, ClO₄⁻, etc.) and phthalocyanines. Although BF₄⁻ and I₃⁻ differ substantially in spatial requirements, single-

(4) (a) Miller, J. S.; Epstein, A. J. Acc. Chem. Res., in press. (b) Epstein,
 A. J.; Miller, J. S.; Pouget, J. P.; Comès, R. Phys. Rev. Lett. 1981, 47,
 741-744. (c) Gutfreund, H.; Little, W. A. In ref 1e, pp 305-372.

(5) (a) Williams, J. M. Prog. Inorg. Chem. 1985, 33, 183-220. (b) Wudl,
F. Acc. Chem. Res. 1984, 17, 227-232. (c) Williams, J. M.; Beno, M. A.;
Sullivan, J. C.; Banovetz, L. M.; Braam, J. M.; Blackman, G. S.; Carlson, C. D.; Greer, D. L.; Loesing, D. M. J. Am. Chem. Soc. 1983, 105, 643-645. (d)
Whangbo, M.-H.; Williams, J. M.; Beno, M. A.; Dorfman, J. R. J. Am. Chem. Soc. 1983, 105, 646-648. (e) Beckgaard, K.; Jerôme, D. Sci. Am 1982, 247, 52-61. (f) Jerôme, D.; Schulz, H. J. Adv. Phys. 1982, 31, 299-490.

(6) (a) Marks, T. J. Science (Washington, D.C.) 1985, 227, 881-889. (b) Inabe, T.; Marks, T. J.; Burton, R. L.; Lyding, J. W.; McCarthy, W. J.; Kannewurf, C. R.; Reisner, G. M.; Herbstein, F. H. Solid State Commun. 1985, 54, 501-503 and references therein. (c) Inabe, T.; Liang, W.-B.; Lomax, J. F.; Nakamura, S.; Lyding, J. W.; McCarthy, W. J.; Carr, S. H.; Kannewurf, C. R.; Marks, T. J. Synth. Met., in press. (d) Inabe, T.; Marks, T. J.; Lyding, J. W.; Imaeda, K.-I.; McCarthy, W. J.; Kannewurf, C. R., submitted for publication. (e) Palmer, S. M.; Ogawa, M. Y.; Martinsen, J.; Stanton, J. L.; Hoffman, B. M.; Ibers, J. A.; Green, R. L. Mol. Cryst. Liq. Cryst. 1985, 120, 427-432 and references therein. (f) Diel, B. N.; Inabe, T.; Lyding, J. W.; Schoch, K. F., Jr.; Kannewurf, C. R.; Marks, T. J. J. Am. Chem. Soc. 1983, 105, 1551-1567. (g) Schramm, C. S.; Scaringe, R. P.; Stojakovic, D. R.; Hoffman, B. M.; Ibers, J. A.; Marks, T. J. J. Am. Chem. Soc. 1980, 102, 6702-6713.

[†] Present address: Institute for Molecular Science, Okazaki, Japan.

^{(1) (}a) Pecile, C., Zerbi, G., Bozio, R., Girlando, A., Eds. "Proceedings of the International Conference on the Physics and Chemistry of Low-Dimensional Synthetic Metals (ICSM 84)"; Abano Terme, Italy, June 17-22, 1984; *Mol. Cryst. Liq. Cryst.* **1985**, 117-121. (b) Miller, J. S., Ed. "Extended Linear Chain Compounds"; Plenum Press: New York, 1982; Vols. 1-3. (c) Epstein, A. J.; Conwell, E. M., Eds. "Proceedings of the International Conference on Low-Dimensional Conductors"; Boulder, CO, Aug 9-14, 1981; *Mol. Cryst. Liq. Cryst.* **1981-1982**, 77, 79, 81, 83, 85, 86, Parts A-F. (d) Alcacer, L., Ed. "The Physics and Chemistry of Low-Dimensional Solids"; D. Reidel: Dordrecht, 1980. (e) Devreese, J. T., Evrard, R. P., van Doren, V. E., Eds. "Highly Conducting One-Dimensional Solids"; Plenum Press: New York, 1979.

^{(2) (}a) Marks, T. J.; Kalina, D. W. In ref 1b, Vol. 1, pp 197-331. (b) Delhaes, P. *Mol. Cryst. Liq. Cryst.* 1983, 96, 229-262. (c) Torrance, J. B. Acc. Chem. Res. 1979, 12, 79-86.

^{(3) (}a) Wiygul, F. M.; Metzger, R. M.; Kistenmacher, T. J. Mol. Cryst. Liq. Cryst. 1984, 107, 115-131 and references therein. (b) Metzger, R. M. J. Chem. Phys. 1981, 75, 3087-1096. (c) Torrance, J. B.; Silverman, B. D. Phys. Rev. B 1977, 15, 788-801. (d) Epstein, A. J.; Lipari, N. O.; Sandman, D. J.; Nielsen, P. Phys. Rev. B 1976, 13, 1569-1579.

<sup>6702-6713.
(7)</sup> Structure-enforced polymeric systems have greater flexibility in this regard.^{6a} (a) Inabe, T.; Moguel, M. K.; Marks, T. J.; Burton, R.; Lyding, J. W.; Kannewurf, C. R. *Mol. Cryst. Liq. Cryst.* 1985, 118, 349-352. (b) Inabe, T.; Lyding, J. W.; Moguel, M. K.; Marks, T. J. J. *Phys. Colloq.* 1983, 3, 625-631. (c) Inabe, T.; Lyding, J. W.; Gaudiello, J. G.; McCarthy, W. J.; Moguel, M. K.; Kannewurf, C. R.; Marks, T. J. J. unpublished results.
(2) Exa recent elactrophemical curthers of other low dimensional mater.

⁽⁸⁾ For recent electrochemical syntheses of other low-dimensional materials, see refs 1a and 5a.

⁽⁹⁾ Preliminary report on powder samples prepared by chemical oxidation: Inabe, T.; Lyding, J. W.; Moguel, M. K.; Kannewurf, C. R.; Marks, T. J. Mol. Cryst. Liq. Cryst. 1983, 93, 355-367.

^{(10) (}a) Inabe, T.; Almeida, M.; Marks, T. J., research in progress. (b) Inabe, T.; Reisner, G. M.; Herbstein, F. H.; Marks, T. J., manuscript in preparation.

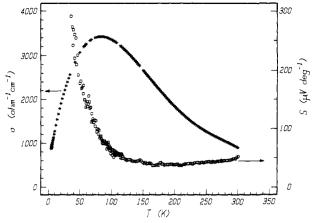


Figure 1. Variable-temperature charge-transport data for typical Ni- $(Pc)(BF_4)_{0.33}$ crystals measured in the macrocycle stacking direction. * Four-probe dc electrical conductivity; \Box thermopower.

crystal diffraction data (300 K) indicate that Ni(Pc)(BF₄)_{0.33} is structurally (P4/mcc, a = 13.97 (2) Å, c = 6.48 (1) Å)^{10b} rather similar to H₂(Pc)I (2, P4/mcc, a = 13.979 (6) Å, c = 6.502 (3) Å)^{6b} and Ni(Pc)I (3, P4/mcc, a = 13.936(6), c = 6.488(3) Å).^{6g} This implies a crystal structure consisting of Ni(Pc)^{+0.33} ions stacked in columns at 3.240 (5) Å separations and surrounded by BF₄⁻ counterions. Room temperature powder EPR measurements confirm the ligand-centered character of the oxidation (g = 2.000) and the unidimensional nature of the electronic structure (line width = 1.1 G).

Despite the structural similarities, the electrical properties of 1 differ significantly from those of $H_2(Pc)I^{6b,c}$ and Ni(Pc)I.^{6e,f} Over 100 crystals were measured by four-probe dc techniques^{6b,11a} using automated instrumentation described elsewhere.^{6f} Appropriate precautions were taken to avoid thermal shock.^{6b} Thermoelectric power measurements were performed by using a conventional type of apparatus.^{11b} As can be seen in Figure 1, typical $Ni(Pc)(BF_4)_{0.33}$ crystals exhibit room temperature conductivity in the macrocycle stacking direction of ca. 1000 Ω^{-1} cm⁻¹—comparable to or higher than that of any known phthalocyanine conductor. Interestingly, and unlike 2 and 3, the "metallike" $(d\sigma/dT < 0)$ temperature dependence of 1 is not sharply peaked but rather goes through a broad metal-to-semiconductor transition at relatively high temperature ($T_{\rm max} \sim 80$ K). While σ_{max} is comparable to that of very pure $H_2(Pc)I$ crystals (3000-4000 Ω^{-1} cm⁻¹),^{6b} the conductivity maximum in the latter occurs at ca. 15 K, with a transition to a semiconductor below.^{6b-d} Even at 1.5 K, the conductivity of $Ni(Pc)(BF_4)_{0,33}$ remains in the 300-600 Ω^{-1} cm⁻¹ range.

At $T \gtrsim 150$ K, the thermopower of Ni(Pc)(BF₄)_{0.33} crystals (Figure 1) evidences "metallike" (σ approximately linear with Tbut less so than for 2 and 3) hole conduction (S > 0), implying transport primarily if not exclusively through the Ni(Pc)^{+0.33} stacks. The magnitude of S (300 K) and the extrapolated nonzero intercept at T = 0 are typical of many radical cation conductors.^{1,2a,5e,6c,12} Assuming a simple tight-binding band model, thermopower can be related to the bandwidth (4t = 4 times the Hückel β integral for ring-ring interaction¹³) via eq 1,¹² where

$$S = \frac{2\pi^2 k_{\rm B}^2 T \cos(\pi \rho/2)}{3e(4t) \sin^2(\pi \rho/2)}$$
(1)

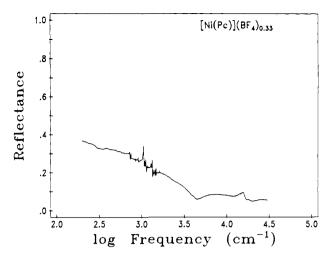


Figure 2. Optical reflectance spectrum of a polycrystalline $Ni(Pc)-(BF_4)_{0.33}$ sample.

 ρ is the degree of partial oxidation.¹² Linear regression analysis of the *S*,*T* data in the high temperature, quasi-linear region (>200 K) yields 4t = 1.3 (1) eV. Below ca. 100 K, the thermopower data reveal that Ni(Pc)(BF₄)_{0.33} undergoes a gradual transition to a semiconductor (cf., TTT₂I₃^{12b,c}).

The specular reflectance of Ni(Pc)(BF₄)_{0.33} samples exhibits a typical "molecular metal"^{1,5e} plasma edge in the infrared (Figure 2). Analysis of the line shape^{6b,f} using a Drude model for the dielectric function¹⁴ and procedures described elsewhere^{6b,f,15} yields the following parameters for the plasma frequency, the electronic relaxation time, and the high frequency dielectric constant: ω_p = 6020 (30) cm⁻¹, τ = 2.49 (5) × 10⁻¹⁵ s, ϵ_{core} = 2.46 (3). These values are similar to those for 2^{6b} and 3.^{6f} Furthermore, ω_p can be related to the tight-binding bandwidth via eq 2, where c is the

$$4t = \frac{\rho(h\omega_{\rm p})^2}{4N_{\rm c}e^2c^2\sin{(\pi\rho/2)}}$$
(2)

interplanar spacing and N_c is the carrier density per unit volume.^{6f} The optically derived bandwidth of Ni(Pc)(BF₄)_{0.33} is found to be 1.2 (1) eV, in excellent agreement with the thermopower result as well as with $4t_{optical} = 1.3$ (1) eV for 2^{6b} and 1.0 (1) eV for 3.^{6f} The weak paramagnetic susceptibility of 1, measured with a SQUID susceptometer, is also reminiscent of the triiodides: $\chi_s = 1.30$ (4) × 10⁻⁴ emu mol⁻¹ and, with the exception of slight "Curie tailing" (attributable to defects, impurities, disorder^{6b,e,17}) at low temperatures, is virtually temperature-independent. In comparison, $\chi_s(2) = 2.21$ (5) × 10⁻⁴ emu mol⁻¹ ^{6b} and $\chi_s(3) = 1.90$ (10) × 10⁻⁴ emu mol⁻¹,^{6f} with similar temperature dependencies. For 1, we calculate a tight-binding "magnetic" bandwidth^{6b,f} of 0.64 (3) eV, the credibility of which likely reflects electron–electron (U)^{16,18} and/or electron–phonon¹⁹ interactions.

 ^{(11) (}a) Shafer, D. E.; Wudl, F.; Thomas, G. A.; Ferraris, J. P.; Cowan,
 D. O. Solid State Commun. 1974, 14, 347-351. (b) Chaikin, P. M.; Kwak,
 J. F. Rev. Sci. Instrum. 1975, 46, 218-220.

^{(12) (}a) Maaroufi, A.; Coulon, C.; Flandrois, S.; Delhaes, P.; Mortenson, K.; Beckgaard, K. Solid State Commun. 1983, 48, 555-559 and references therein. (b) Khanna, S. K.; Fuller, W. W.; Chaikin, P. M. Phys. Rev. B 1981, 24, 2958-2963. (c) Chaikin, P. M.; Grüner, G.; Schegolev, I. F.; Yagubskii, E. B. Solid State Commun. 1979, 32, 1211-1214. (d) Schweitzer, D.; Hennig, I.; Bender, K.; Endres, H.; Keller, H. J. Mol. Cryst. Liq. Cryst. 1985, 120, 213-220. (e) Mortensen, K.; Jacobsen, C. S.; Bechgaard, K.; Carneiro, K.; Williams, J. M. Mol. Cryst. Liq. Cryst. 1985, 119, 401-404.

^{(13) (}a) Pietro, W. J.; Ratner, M. A.; Marks, T. J. J. Am. Chem. Soc.
1985, 107, 5387. (b) Ciliberto, E.; Doris, K. A.; Pietro, W. J.; Reisner, G. M.; Ellis, D. E.; Fragalá, I.; Herbstein, F. H.; Ratner, M. A.; Marks, T. J. J. Am. Chem. Soc. 1984, 106, 7748-7761. (c) Pietro, W. J.; Ellis, D. E.; Marks, T. J.; Ratner, M. A. Mol. Cryst. Lig. Cryst. 1984, 105, 273-287. (14) Tanner, D. B. In ref 1b, Vol. 2, pp 205-258.

^{(15) (}a) These techniques allow reliable intercomparison of optical data from single-crystal and polycrystalline samples. (b) McCarthy, W. J.; Kannewurf, C. R.; Inabe, T.; Marks, T. J.; Burton, R. L. NBS Spec. Publ. (U.S.) 1985, No. 697, 54-57.

 ^{(16) (}a) Shiba, H. Phys. Rev. B 1972, 6, 930-938. (b) Torrance, J. B.;
 Tomkiewicz, Y.; Silverman, B. D. Phys. Rev. B 1977, 15, 4738-4749.
 (17) (a) Delhaes, P.; Coulon, C.; Flandrois, S.; Hilti, B.; Mayer, C. W.;

^{(17) (}a) Delhaes, P.; Coulon, C.; Flandrois, S.; Hilti, B.; Mayer, C. W.;
Rihs, G.; Rivory, J. J. Chem. Phys. 1980, 73, 1452-1463. (b) Tippie, L. C.;
Clark, W. G. Phys. Rev. B 1981, 23, 5846-5853. (c) Soos, Z. G.; Bondeson,
S. R. Mol. Cryst. Liq. Cryst. 1982, 85, 19-31. (d) Takahashi, M.; Sugano,
T.; Kinoshita, M. Bull. Chem. Soc. Jpn. 1984, 57, 26-35.

 ^{(18) (}a) Delhaes, P. Mol. Cryst. Liq. Cryst. 1983, 96, 229-262. (b)
 Torrance, J. B.; Tomkiewicz, Y.; Bozio, R.; Pecile, C. R.; Wolfe, C. R.;
 Bechgaard, K. Phys. Rev. B 1982, 26, 2267-2270. (c) Mortensen, K.; Conwell, E. M.; Fabre, J. M. Phys. Rev. B 1983, 28, 5856-5862. (d) Mazumdar,
 S.; Bloch, A. N. Phys. Rev. Lett. 1983, 50, 207-211.

These results indicate that phthalocyanine "molecular metal" chemistry is considerably broader than that involving only halogen counterions. In the case of highly conductive $Ni(Pc)(BF_4)_{0.33}$, similarities to halogenated analogues in crystal and band structure are accompanied by significant differences in transport/metal-to-semiconductor transition characteristics. Further phthalocyanine counterion effects are under investigation.

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(19) Gutfreund, H.; Entin-Wohlman, O.; Weger, M. Mol. Cryst. Liq. Cryst. 1985, 119, 457-466 and references therein.

Organometallic Clusters Containing Oxygen Atoms: Preparation, Structure, and Properties of $(\eta$ -C₅H₅)₁₁V₁₃O₁₈(N(CH₃)₃)₂ and $(\eta$ -C₅H₅)₁₄V₁₆O₂₄, Dimeric Derivatives of $(\eta$ -C₅H₅)₆V₆O₈

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We described previously $Cp_5V_5O_6$ and $Cp_4Cr_4O_4^{-1}$ ($Cp = \eta^5 - C_5H_5$) which with $Cp_6Ti_6O_8$ described by Caulton et al.² form a series of $(CpM)_m(\mu_3-O)_n$ clusters. In a theoretical study we suggested that in each cluster there were 12 orbitals occupied by 2 ($Cp_6Ti_6O_8$), 8 ($Cp_5V_6O_6$), or 12 ($Cp_4Cr_4O_4$) electrons, and a variety of more or less distorted octahedral (CpM)₆O₈, trigonal bipyramidal (CpM)₅O₆, and tetrahedral (CpM)₄O₄ clusters should be obtainable.³ We report here the preparation of $Cp_6V_6O_8$ and two remarkable derivatives of it, $[(Cp_5V_6(\mu_3-O)_8)_2((\mu-O)_2VCp-(N(CH_3)_3)_2)]$ (I) and $[(Cp_5V_6(\mu_3-O)_8)_2(CpV)_4(\mu_2-O)_8]$ (II).

When Cp₂V was oxidized by (CH₃)₃NO in toluene a black solid and a black-brown solution were obtained. The solid was not crystalline but mass spectrometry (all m/e peaks corresponding to Cp_mV₆O₈⁺ with $m = 6 \rightarrow 0$), microanalysis (found, C, 42.5; H, 3.9%;, calcd for C₃₀H₃₀V₆O₈, C, 43.7; H, 3.7%) and ¹H NMR (singlet at -171.4 ppm) established it as paramagnetic Cp₆V₆O₈.

The solution deposited large crystals on storage at 5 °C. These were shown to be I by X-ray crystallography (Figure 1).⁴ The cluster is derived from Cp₆V₆O₈ by removal of one Cp and linking of two Cp₅V₆O₈ fragments via the oxygen of (CpV(N(CH₃)₃)₂-(μ -O)₂). The Cp₅V₆O₈ units contain regular octahedra of vanadium with an average V–V distance of 2.906 (4, 29) Å.⁵ The oxygen atoms lie over the triangular faces with an average V–O distance of 1.940 (10, 55) Å. The V–V distances are similar to the Ti–Ti in Cp₆Ti₆O₈, 2.891 Å (1, 14).² The V–O distance to the linking oxygen is 1.679 (15) Å.

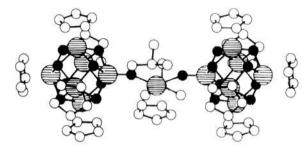


Figure 1. Structure of $(\eta - C_5 H_5)_{11} V_{13} O_{18} (N(CH_3)_3)_2$.

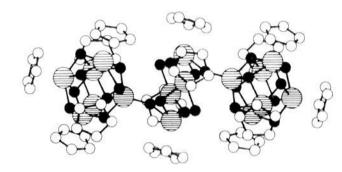


Figure 2. Structure of $(\eta - C_5H_5)_{14}V_{16}O_{24}$.

The ESR of I showed an assymptrical eight-line spectrum typical of vanadium in a low-symmetry environment. The ¹H NMR showed a broad, weak, symmetrical resonance at -79.4, a multiplet at -7.15, and a singlet at -2.29 ppm. The latter resonances had an intensity ratio of 2.6:1. We assign the resonance at -79.4 ppm to an η^5 -C₅H₅ group attached to paramagnetic vanadium, the resonance at -7.15 ppm to η^5 -C₅H₅ attached to diamagnetic vanadium, and that at -2.29 ppm to the N(CH₃)₃. This interpretation means that the Cp₅V₆O₈ units are diamagnetic and the vanadium(III) in the bridge has two unpaired electrons.

The diamagnetism and regularity of the Cp₅V₆O₈ may be rationalized by arguments developed for Cp₆M₆A₈.³ The Cp₅(μ -O)V₆O₈ units are derivatives of Cp₆V₆O₈ in which an OR group replaces one Cp. This leaves the number of cluster electrons at eight but increases the cluster orbitals by two. If the unique vanadium lies on z the four orbitals it contributes to the cluster are the same two as in Cp₆V₆O₈, d_{x²-y²} and d_{z²}, plus d_{xz} and d_{yz}. Of these, d_{x²-y²} is nonbonding and the other three are antibonding counterparts of V-O bonds. In addition d_{xz} is the antibonding counterpart of the π -bond between the bridging oxygen and vanadium. Therefore d_{x²-y²} remains localized on the unique vanadium and is occupied by two electrons; d_{z²} and d_{yz} contribute to the cluster orbitals. The configuration is a_{1g}²e_g⁴; therefore Cp₅V₆O₈ is diamagnetic and undistorted.

When Cp₂V was oxidized by $(CH_3)_3NO$ in tetrahydrofuran a black, diamagnetic, crystalline solid of formula Cp₁₄V₁₆O₂₄ was obtained. It too is derived from Cp₆V₆O₈: two Cp₅V₆O₈ are linked by $(CpV)_4(\mu_2-O)_8$ (Figure 2).⁶ The latter is a rectangle of vanadium with pairs of oxygen along the edges; the connections to Cp₅V₆O₈ are diagonally opposed oxygens (V–O 1.701 (13) Å). The V–V distances in the Cp₅V₆O₈ units average 2.900 (5, 21) and the V–O distances 1.942 (14, 67) Å, very similar to I. In the $(CpV)_4(\mu_2-O)_8$ bridge the V–V distances are 2.989 (6) and 3.296 (6) Å and the V–O distance to the doubly bridging oxygens 2.020 (14, 29) Å but to the two triply bridging ones 2.179 (14, 26).

These clusters are extreme cases of those obtained by binding organometallic fragments to polyoxometalate ions;⁷⁻⁹ {[(C_7H_8)-Rh]₅(*cis*-Nb₂W₄O₁₉)₂}³⁻⁷ is a spectacular example. Both I and II are true molecular clusters, being highly soluble in toluene.

⁽¹⁾ Bottomley, F.; Paez, D. E.; White, P. S. J. Am. Chem. Soc. 1982, 104, 5651.

⁽²⁾ Huffman, J. C.; Stone, J. G.; Krusell, W. C.; Caulton, K. G. J. Am. Chem. Soc. 1977, 99, 5829.

⁽³⁾ Bottomley, F.; Grein, F. Inorg. Chem. 1982, 21, 4170.

⁽⁴⁾ Crystallographic data: $C_{83}H_{105}N_2O_{18}V_{13}$ ((C_5H_5)₁₁ $V_{13}O_{18}$ (N(C-H₃)₃)₂·4($C_6H_5CH_3$)), $M_r = 2081.12$, C2/m, a = 17.969 (4) Å, b = 14.318 (3) Å, c = 18.853 (4) Å, $\beta = 105.94$ (28)°; Z = 2; $D_c = 1.48$ g cm⁻³; R = 0.095, $R_w = 0.108$ for 1674 observed ($I > 3\sigma(I)$) reflections with $2\theta < 45^\circ$. Anisotropic V and O; the N(CH₃)₃ and C₅H₅ groups were refined as rigid bodies (137 variables). There is disorder in the CpV(μ -O)₂(N(CH₃)₃)₂ unit and in the toluenes.

⁽⁵⁾ The first figure is the average standard deviation, the second the maximum deviation from the average.

⁽⁶⁾ Crystallographic data: $C_{77}H_{78}O_{24}V_{16} ((C_5H_5)_{14}V_{16}O_{24}\cdot C_6H_5CH_3), M_r = 2202.5, P\bar{1}, a = 11.582$ (2) Å, b = 13.099 (2) Å, c = 13.969 (4) Å, $\alpha = 94.10$ (2)°, $\beta = 90.03$ (2)°, $\gamma = 104.72$ (2)°, Z = 1; $D_c = 1.78$ g cm⁻³; $R = 0.070, R_w = 0.076$ for 1657 observed ($I > 3\sigma(I)$) reflections with $2\theta < 45^\circ$. Anisotropic V and O; C_5H_5 groups as rigid bodies (287 variables).

⁽⁷⁾ Besecker, C. J.; Klemperer, W. G.; Day, V. W. J. Am. Chem. Soc. 1982, 104, 6158.

⁽⁸⁾ Knoth, W. H. J. Am. Chem. Soc. 1979, 101, 2211.

⁽⁹⁾ Finke, G. R.; Droege, M. W.; Cook, J. C.; Suslick, K. S. J. Am. Chem. Soc. 1984, 106, 5750.